

Increasing the photocatalytic efficiency and electro-optical properties of titanium dioxide nanolayers doped with copper, silver and gold elements

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Abstract: To improve the photocatalytic activity of titanium dioxide nanolayers, we used doping with gold, silver, and copper nanoparticles. The nanolayers were prepared by the pulsed laser deposition (PLD) method. The surface morphology of the nanolayers was investigated using scanning electron microscopy (SEM), and the roughness and surface properties of the samples were measured using atomic force microscopy (AFM). X-ray diffraction (XRD) was used to characterize the crystal structure of the nanolayers. Also, the absorption spectra of the nanolayers were measured using a UV-Visible spectrophotometer. Their absorption and band gap changes were investigated and compared. A high increase in the absorption rate of nanolayers, especially compared to previous studies, was observed in the obtained results. The photocatalytic activity of the nanolayers was analyzed using the ISO-10678:2010 method for photonic efficiency, and the effect of doped elements in the structure of the nanolayers was studied. The results showed that doping titanium dioxide with metallic elements such as copper, silver, and gold significantly improves the photonic efficiency of nanolayers.

Keywords: titanium dioxide; pulsed laser deposition; nanoparticles; absorption efficiency; photonic efficiency.

1. Introduction

More than 2000 semiconductor materials have now been identified by researchers, most of which are oxides, sulfides, or nitrides based on transition metals [1–3]. Titanium dioxide remains one of the most promising semiconductor materials because of its high efficiency, low cost, chemical inertness, eco-friendly nature and Photo-stability [4,5].

Improving the electro-optical properties and increasing the efficiency of photocatalytic activity of photocatalytic materials and thin films is a hot topic for many researchers [6]. In a photocatalytic reaction in a semiconductor, upon irradiation with light, electrons in the valence band gain the energy required to be excited and jump to the conduction band, creating an electron-hole pair in the conduction and valence bands, respectively [7–16]. The electron-hole pair produced by the light migrates to the surface of the semiconductor material and participates in the photocatalytic reactions in the degradation of pollutants [17–21]. It should also be noted that the light absorption range of semiconductor materials is the visible light region up to a wavelength of 700 nm, making them very suitable for absorbing radiant energy from sunlight.

Doping nanolayers with elements that accelerate charge transfer reactions and also reduce the electron-hole pair recombination process is one way to increase the efficiency of photocatalytic activity, which can also lead to improvement of their

electro-optical properties. Titanium dioxide has also been and is the target of many researchers as a photocatalytic material whose photocatalytic activity is mainly in the ultraviolet radiation range [22–24].

Many efforts have been focused on reducing the optical band gap of titanium dioxide in order to improve its electro-optical properties. Therefore, doping of titanium dioxide nanolayers with various metal nanoparticles and their effect on electro-optical properties and photocatalytic activity have been investigated and studied [25–28]. Doping titanium dioxide with metal ions leads to a significant increase in photocatalytic activity, especially when it is doped with silver and copper, which has attracted considerable attention [26–28].

Varieties of nanocomposite semiconductor materials have been synthesized in recent years to improve the selectivity and efficiency of photocatalytic processes. One of the main concerns in photocatalytic activity is to improve their absorption coefficient. Therefore, doping different metal nanoparticles of different sizes, each of which absorbs different wavelengths of incident light, is one way to achieve this goal. For example, the effect of silver nanoparticles on increasing the absorption coefficient of TiO₂ thin films can be mentioned [29–34]. Another study investigated the effects of doping titanium dioxide nanolayers with copper nanoparticles, which led to a significant increase in the absorption coefficient [35]. It is also possible to mention a study that investigated the effect of gold nanoparticles on increasing the absorption coefficient of titanium dioxide nanolayers [36,37].

Also, many studies have investigated ways to enhance in photocatalytic activity of thin titanium dioxide layers. One of these studies is the high photocatalytic activity and degradation of methyl orange under UV irradiation using silver-doped titanium multilayer membranes by A. Alem and the other is the surface degradation of TiO₂ thin films modified with silver nanoparticles under sunlight irradiation by Ibrahim H.M.M and also the enhancement of photocatalytic activity of TiO₂ on Ag substrate by Nyamukamba et al. [38–40]. Also, the author has previously conducted studies on titanium dioxide nanolayers deposited on copper and silver nanolayers as well as on copper and silver nanoclusters [41–46]. Many studies have been conducted in this field and some interesting research can be found in these references [47–50].

In this research, the aim is to increase the photocatalytic efficiency of titanium dioxide nanolayers as well as titanium dioxide nanolayers doped with copper, silver and gold and compare them. In addition, the effect of UV irradiation time on the photocatalytic efficiency has been investigated.

2. Materials and methods

2.1. Experimental methods

Each time, one of the samples is placed in the vacuum chamber as a target. The vacuum system was pumped from 10⁻⁶ Torr to a final pressure of 2 × 10⁻⁵ Torr. The substrates were made of quartz and cut into 10 × 10 mm pieces. Pulsed laser deposition (PLD) in the presence of argon gas was used to deposit pure and doped TiO₂ nanolayers. A KrF laser (248 nm wavelength, 2 J/cm² energy density, 7.5 × 10⁻³

Torr argon gas, 25 °C substrate temperature, 5 cm target-substrate distance, 10 Hz pulse repetition rate, and 10,000 pulses) was used to grow TiO₂ nanolayers. Since the energy required to detach atoms from the target surface is provided by momentum transfer, the impurity or purity of the target is not important. The titanium dioxide powder that was used in this work was prepared by Kimia Eksir Company. Copper, silver, and gold powders were used as impurities. The dopant is added to titanium dioxide powder in 5% the weight, as **Table 1** and four different samples were prepared, and were initially completely cleaned of contaminants.

Table 1. Weight percentage of dopant in different samples.

Sample	Titanium dioxide (%)	Dopant (%)
S ₁	100	0
S ₂	95	Cu (5%)
S ₃	95	Ag (5%)
S ₄	95	Au (5%)

Then, nanolayers were deposited by sputtering and their thickness was measured during the deposition process. The nanolayers were coated on a quartz substrate with a thickness of 350 ± 35 nm.

2.2. Analysis of samples

The X-ray diffraction patterns of the nanolayers were prepared using a PHILIPS model PW1730 (Cu α with a wavelength of 0.154056 nm). The surface morphology of the nanolayers was studied using a Hitachi S-4160 field emission electron microscope (FE-SEM). The roughness and surface properties of the samples were measured using image recording using the contact mode of the Auto Probe CP atomic force microscope (AFM) manufactured by Park Scientific Instruments, USA. The absorption spectra of the layers in the range of 200 to 1100 nm were measured using an Ocean optics HR4000 UV-Vis spectrometer.

2.3. Photocatalytic activity

ISO-10678:2010 is a standard for measuring and comparing the photocatalytic activity of different nanolayers. This is done by measuring the photonic efficiency of the nanolayers. This photonic efficiency is equal to the percentage of photons irradiated per unit area that participate in photocatalytic activity per unit time, and in this standard, methylene blue is used as the dye solution and ultraviolet radiation with a wavelength of 365 nm is used [45,46].

Dye solution was added to a cuvette containing a nanolayer with a 1 cm² area. To establish an adsorption and desorption equilibrium, the cuvette was placed in the dark for 24 h, and then it was exposed to ultraviolet light at different times. The radiation source was capable of providing a radiation intensity (E) of 14 W/m² at a distance of 7 cm from the surface of the samples. A UV intensity measuring device (UV-340A) manufactured by LUTRON, Taiwan, was used to measure the radiation intensity of the source. The photonic efficiency can be obtained as follows:

$$\zeta_{MB} = [(\Delta A_\lambda V) / (\Delta t A \varepsilon d \lambda_{max} E)] \times 100 \quad (1)$$

In this equation, λ_{max} and E are the maximum wavelength and the source intensity of radiation, respectively, where E is equal to 14 W/m^2 . Also, ΔA_λ , Δt , V , A , d and ε are the changes in methylene blue absorption, irradiation time, volume of solution under irradiation, active surface area, cuvette irradiation length and methylene blue offset coefficient, respectively, which ($\varepsilon_{664 \text{ nm}} = 7402.8 \text{ m}^2/\text{mol}$). The device used for UV-visible spectroscopy measurements is a Varian spectrometer (model Cary 100 Bio).

3. Results and discussion

3.1. XRD analysis

Figure 1 shows XRD patterns of samples.

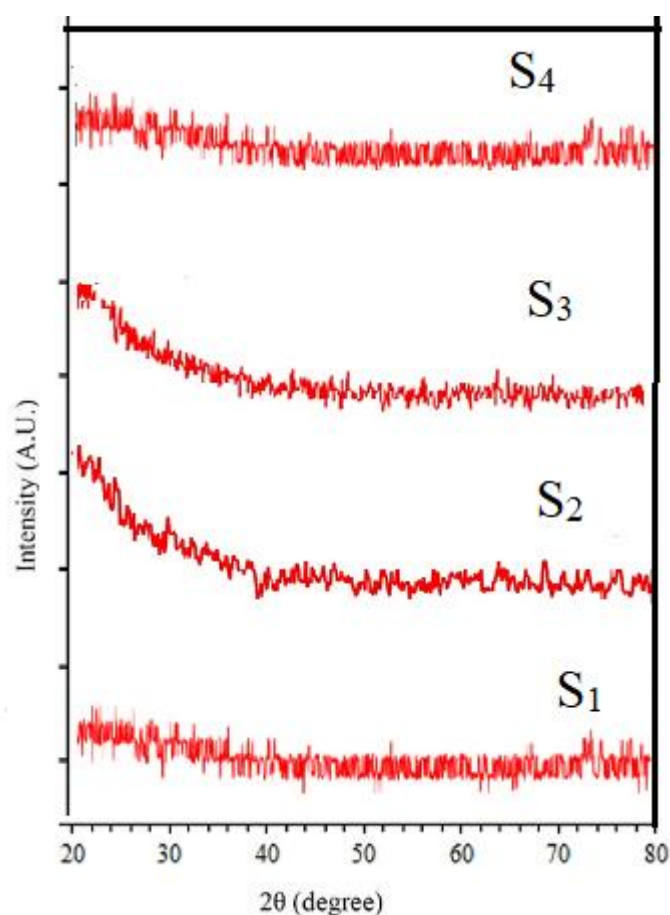


Figure 1. XRD patterns of samples.

Figure 1 shows that all samples, both pure titanium dioxide nanolayers and doped nanolayers, are completely amorphous and lack a crystalline structure. This is in complete agreement with the characteristics of titanium dioxide nanolayers that have not been subjected to heat treatment, and we know that with heat treatment at a temperature of close to $400 \text{ }^\circ\text{C}$, the phase transition process from amorphous to anatase begins and is completed at higher temperatures.

Based on XRD, which shows broad peaks indicating an amorphous nature, it can be concluded that the impurities are present in the form of highly dispersed metal nanoclusters and substitution ions embedded in the TiO_2 structure, which is consistent with the observed electronic behavior.

3.2. Surface morphology of nanolayer

Figures 2 and 3 show the FE-SEM images of the samples, in which we can see that the particle size in all samples is in the range of 10 to 300 nm. However, the particle size in each sample is slightly different. This difference in size could be due to the impurity elements and the difference in their particle size, which causes the formation and orientation of particles with different sizes. Image of sample S_1 (image a) shows a nanolayer with a rough surface and a compact structure. The surface morphology of sample S_2 (image b) shows a rougher and more compact structure with a slightly larger particle size than S_1 . The particle size and surface roughness of the nanolayers in sample S_3 (image c) further increase and become more compact, and finally reach their maximum in sample S_4 (image d).

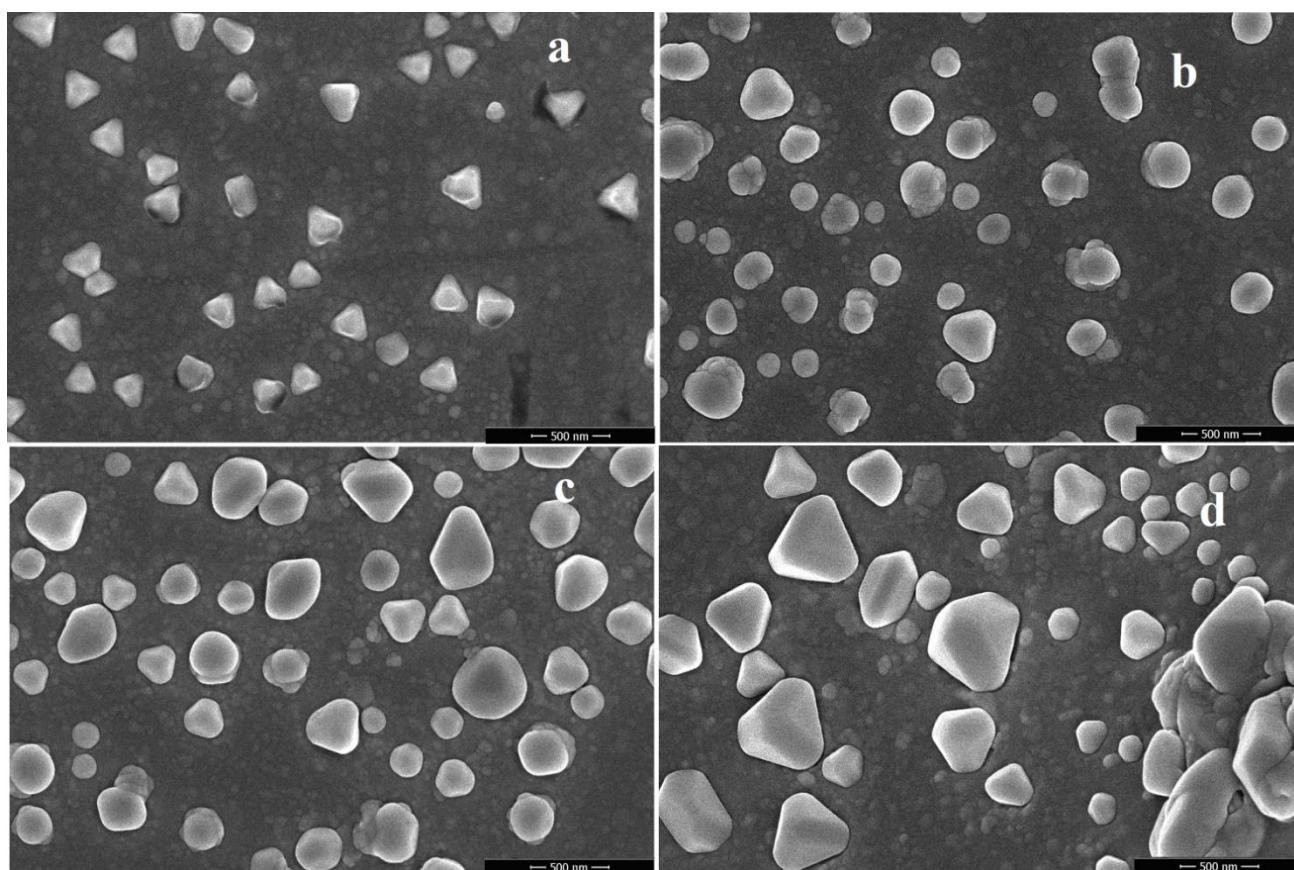


Figure 2. FE-SEM images of samples at 500 nm magnification: (a) S_1 , (b) S_2 , (c) S_3 , and (d) S_4 .

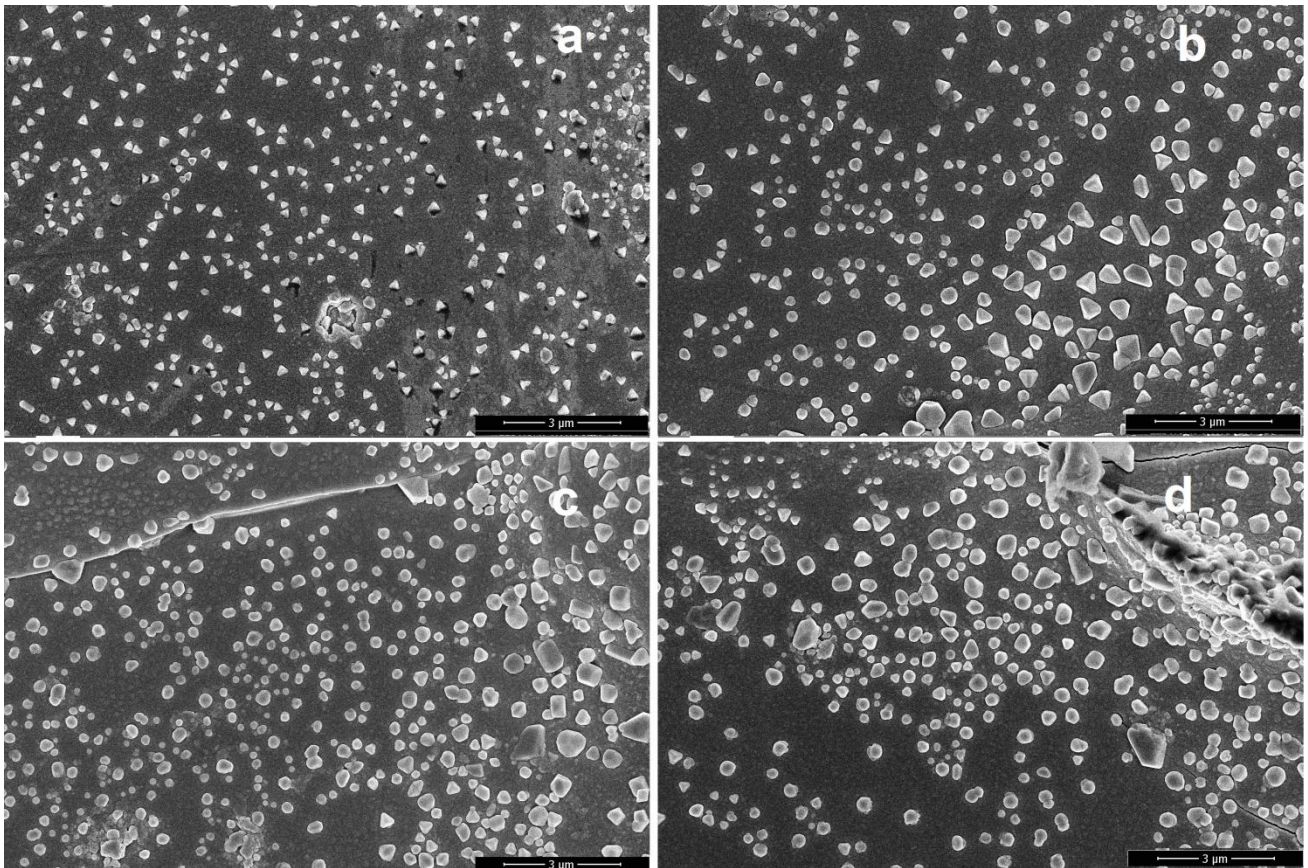


Figure 3. FE-SEM images of samples at 3 μm magnification: (a) S1, (b) S2, (c) S3, and (d) S4.

3.3. AFM analysis

Figure 4 shows the AFM analysis of the samples. The AFM analysis shows an increase in roughness and formation of pores on the surface of the nanolayer in the presence of dopants. This roughness is higher in samples S₂, S₃ and S₄ than in sample S₁. The information on the average linear roughness of the samples obtained from AFM analysis is shown in **Table 2**.

Table 2. Linear roughness average (Ra) of all samples obtained from AFM analysis.

Sample	R _a (nm)
S ₁	0.309
S ₂	0.408
S ₃	0.491
S ₄	0.527

AFM images and linear surface roughness results show that the surface roughness of the nanolayers increased with the presence of metal contaminants such as copper, silver, and gold.

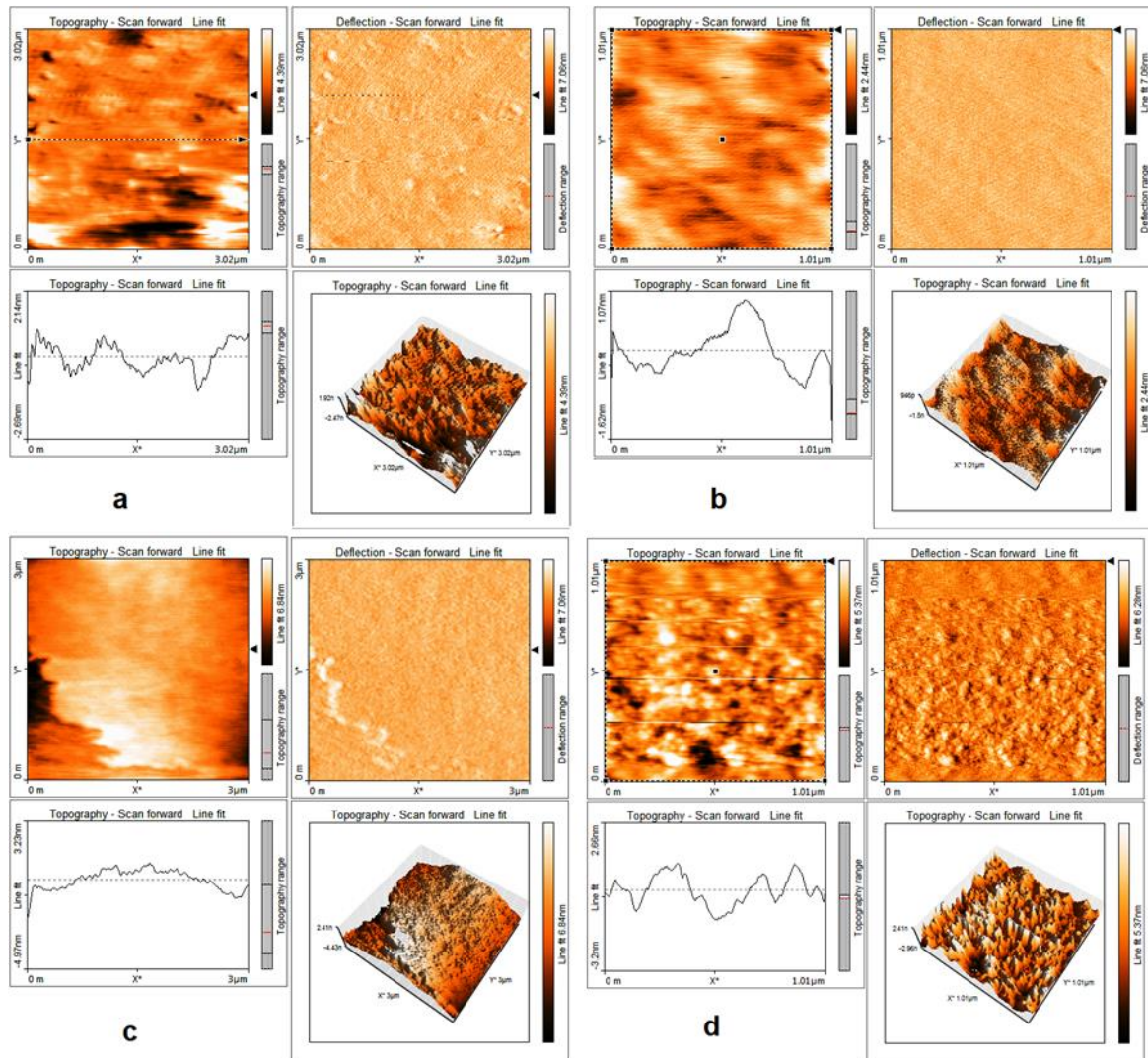


Figure 4. 2D, 3D images and linear topography of AFM analysis: (a) S₁, (b) S₂, (c) S₃, and (d) S₄

3.4. Absorption spectra of the samples

Figure 5 shows the UV-Vis spectra of pure and doped titanium dioxide nanolayers. As can be seen, the absorption coefficient of all samples in the ultraviolet range is high. However, in all samples, as the wavelength increases and approaches the visible region, it decreases sharply, and in the visible region, the absorption is very low and the transmission of the nanolayers is high. The absorption of the samples in the visible region is slightly different, which is due to the doped elements. So, in samples S₃ and S₄, we observe a higher absorption rate, which is due to the presence of gold and silver in the structure of the layers. In addition, in samples S₂ to S₄, several wide absorption bands are observed, and the presence of these absorption bands is more noticeable in sample S₃. The reason for this increase is the vibrations of surface plasmons caused by the presence of copper, silver, and gold nanoparticles on the surface of the nanolayers, which have been mentioned in some articles [43,45,46]. The location, number, and width of these absorption bands vary due to differences in doped elements.

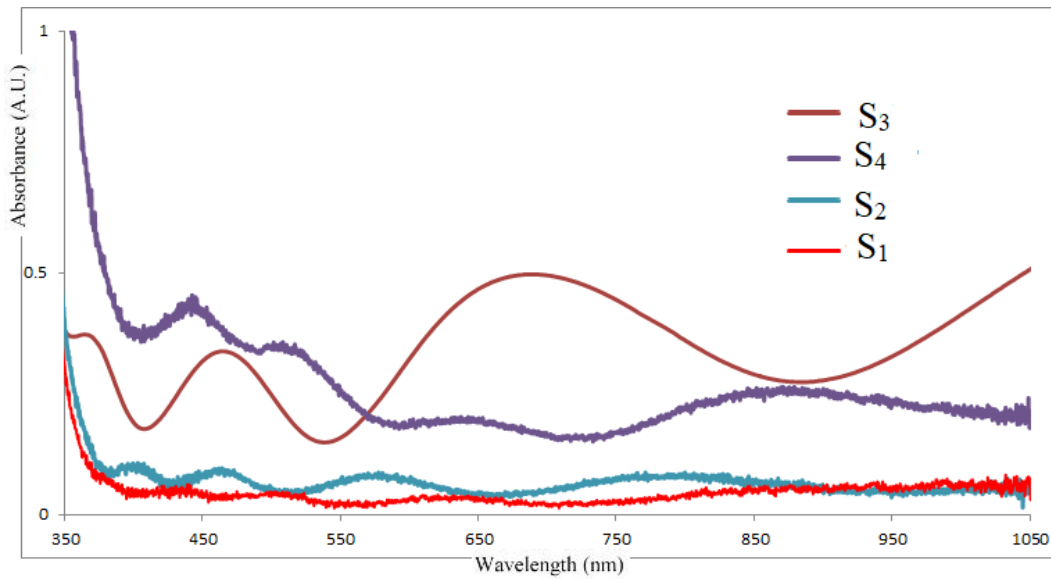


Figure 5. The UV-Vis spectra of samples.

The optical constants of nanolayers, such as the optical absorption coefficient α , can be calculated in various ways. One of the simplest and most popular laboratory methods is to use their absorption and transmission spectra, which are defined as follows, depending on the thickness of the nanolayer:

$$\alpha = \frac{2.303}{t} \log\left(\frac{I_0}{I}\right) \quad (2)$$

In this equation, I_0 , I and t are the intensity of the light incident on the layer, the intensity of the transmitted light and, the layer thickness, respectively. The absorbance edge (E_g) of titanium dioxide nanolayers is shifted to vision region using the Tauc's equation [51]:

$$(\alpha h\nu)^2 = k(h\nu - E_g)^n \quad (3)$$

Where $h\nu$ is the photon energy and k and n are constants, with the value of n being 1 or 3 for semiconductors with direct and indirect energy band gaps, respectively. When Equation (2) is equal to zero, the energy band gap will be equal to the energy of the photon shown. Therefore, the energy band gap of the nanolayer can be calculated from the Tauc's curves. **Figure 6** shows the Tauc's curves for the samples. Where the curves become linear, the direct energy band gaps of these semiconductors can be calculated. The direct energy band gap values of the samples were calculated and listed with an uncertainty of 0.16 eV in **Table 3**.

Table 3. The direct energy band gap of samples.

Sample	S ₁	S ₂	S ₃	S ₄
E_g (eV)	3.19	3.12	2.97	2.86

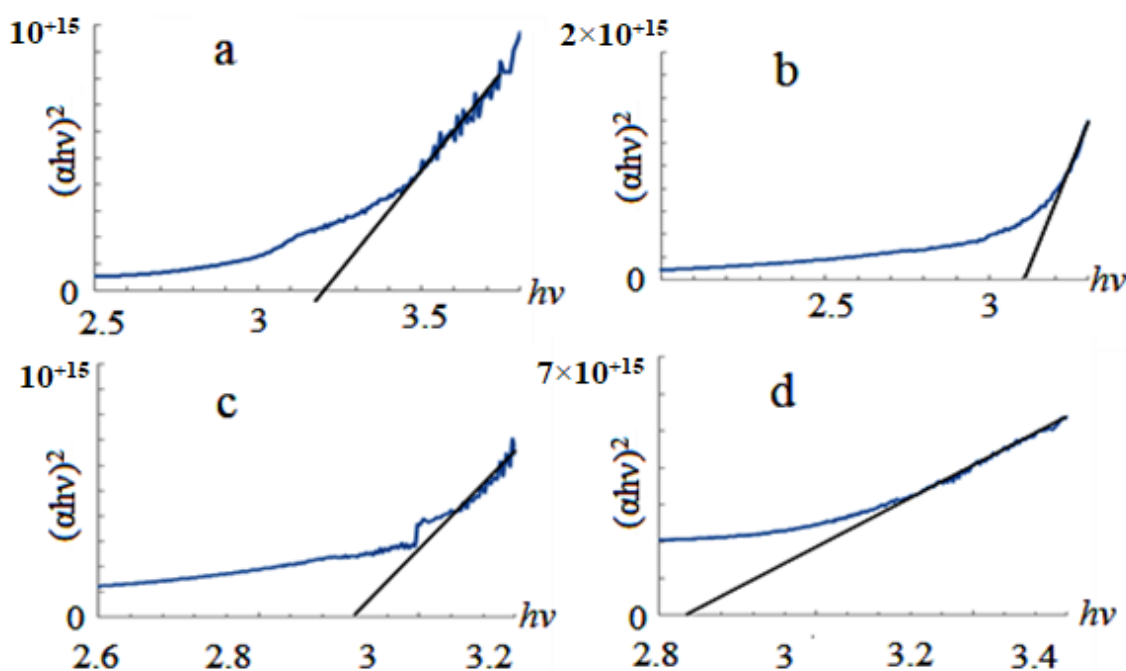


Figure 6. The curves of Tauc's equation of samples.

The reason for the band gap reduction in the presence of metal impurities could be the presence of a metal Fermi level in the band gap of titanium dioxide, which has been reported in some papers [33]. For example, the band gap reduction (from 3.19 eV for S₁ to 2.86 eV for S₄ in **Table 3**) is due to mid-gap defect states originating from localized d-orbitals of the noble metal dopants. **Figure 7** shows a schematic diagram of the energy band structure, showing the interband states derived from the valence band (VB), conduction band (CB), d-orbitals, and band gap for pure and doped titanium dioxide.

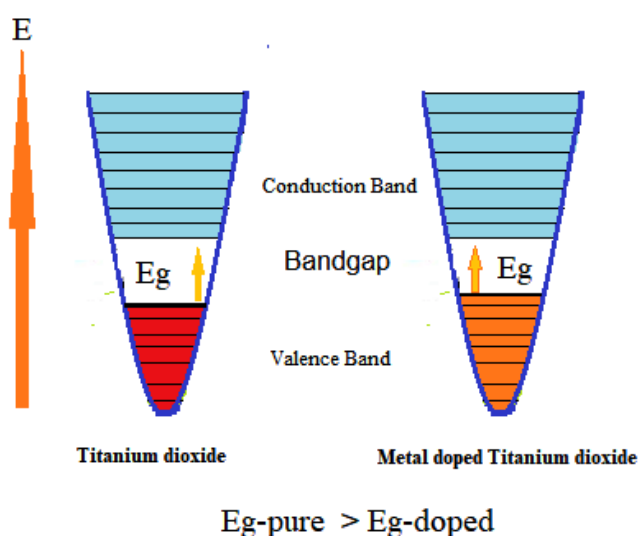


Figure 7. Schematic diagram of the energy band structure of pure and doped titanium dioxide.

3.5. Photocatalytic test

The initial concentration of the Methylene Blue (MB) solution used for all degradation experiments is 2 mL of a 10 μ M solution. Each sample with an active surface area of 1 cm² was placed at the bottom of a cuvette. Then, 2 mL of a 10 μ M methylene blue solution was added to each cuvette. To balance adsorption and desorption between the surface of the layer and the solution, we placed cuvettes containing the samples and the methylene blue solution in the dark for 24 h. A control sample in which the cuvette lacked the nanolayer was also examined. After 24 h, the solution was exposed to UV radiation and samples were taken at 30, 60, 90, and 120 min after the start of irradiation. Spot absorption was measured by ultraviolet-visible spectroscopy at a wavelength of 664 nm. The results obtained are shown in **Table 4**.

Table 4. Spot absorption of samples was measured by ultraviolet-visible spectroscopy.

Sample	A _{30 min}	A _{60 min}	A _{90 min}	A _{120 min}
Control sample	0.5391	0.5388	0.5386	0.5385
S ₁	0.4317	0.3934	0.3413	0.2925
S ₂	0.4174	0.3594	0.3202	0.2747
S ₃	0.4047	0.3517	0.3188	0.2691
S ₄	0.4012	0.3474	0.3106	0.2670

The changes in the absorption coefficient of the samples exposed to ultraviolet radiation after 30, 60, 90, and 120 min from the start of radiation were calculated using **Table 4**, and the results are given in **Table 5**. **Figure 8** also shows the information obtained in **Table 4** for the changes in the absorption coefficient.

Table 6 shows the results of the photonic efficiency under UV irradiation after 30, 60, 90 and 120 min, which are calculated based on the information in **Table 5**. **Figure 9** shows the changes in the photonic efficiency of the samples based on the UV irradiation time.

Table 5. Changes in the absorption coefficient of samples depending on the time of ultraviolet irradiation.

Sample	$\Delta A_{30 \text{ min}}$	$\Delta A_{60 \text{ min}}$	$\Delta A_{90 \text{ min}}$	$\Delta A_{120 \text{ min}}$
S ₁	0.1074	0.1454	0.1973	0.2460
S ₂	0.1217	0.1794	0.2184	0.2638
S ₃	0.1344	0.1871	0.2198	0.2694
S ₄	0.1379	0.1914	0.2280	0.2715

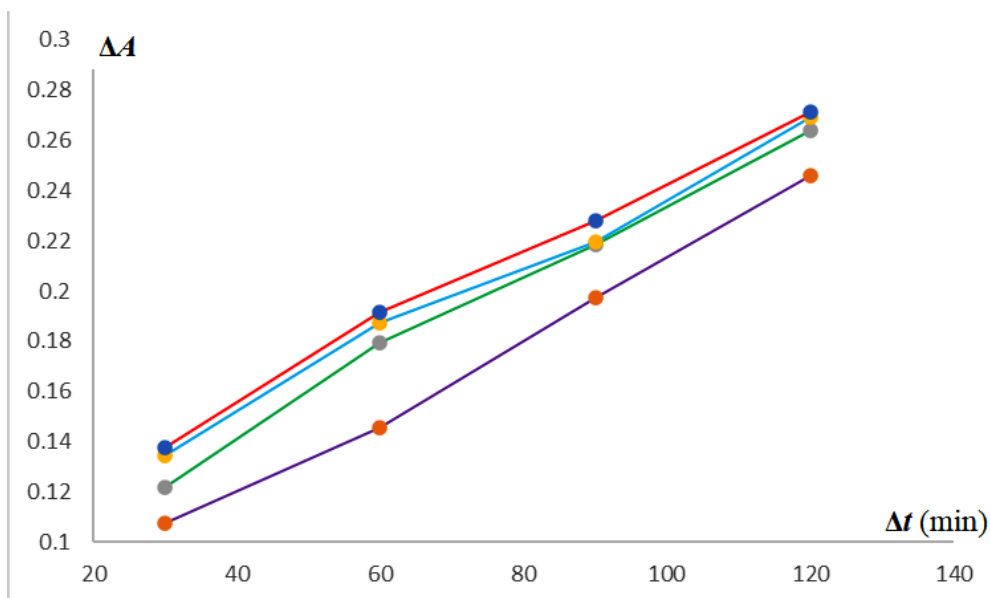


Figure 8. Graph of changes in the absorption coefficient of samples depending on the time of ultraviolet irradiation.

Table 6. The photonic efficiency under UV irradiation after 30, 60, 90 and 120 min.

Sample	$\zeta_{\text{MB-30 min}} (\%)$	$\zeta_{\text{MB-60 min}} (\%)$	$\zeta_{\text{MB-90 min}} (\%)$	$\zeta_{\text{MB-120 min}} (\%)$
S ₁	19.37	13.11	11.86	11.09
S ₂	21.95	16.18	13.13	11.89
S ₃	24.24	16.87	13.21	12.15
S ₄	24.88	17.26	13.71	12.24

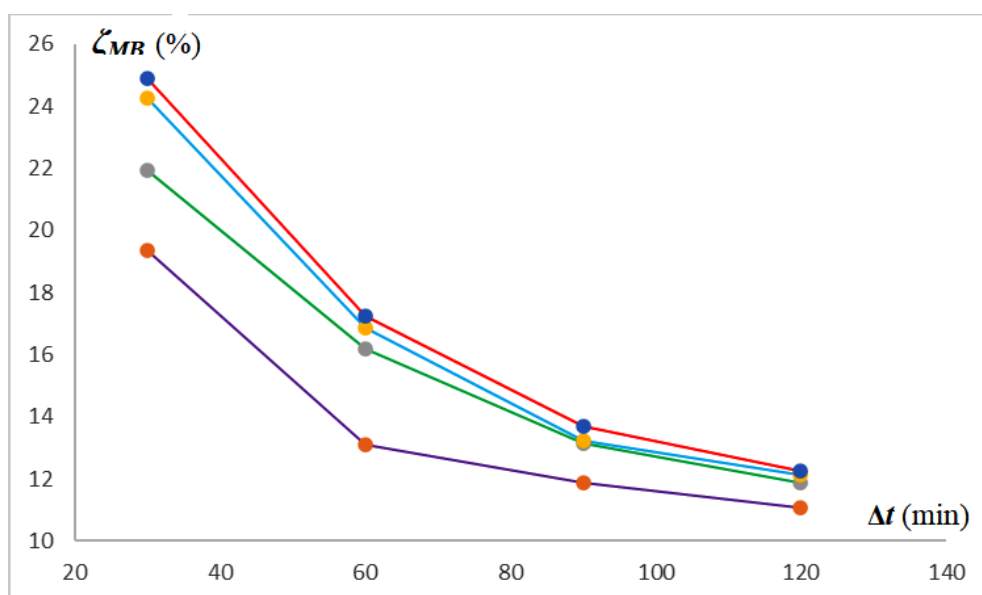


Figure 9. The changes in the photonic efficiency of the samples depending on the time of ultraviolet irradiation.

Doping of metallic elements copper, silver and gold in the structure of titanium dioxide nanolayers due to the difference in particle size increases the surface roughness of the nanolayers, which leads to an increase in surface active sites in

photocatalytic activity and the surface charge carrier transfer ratio. Therefore, the absorption coefficient and photonic efficiency of the nanolayers increase [22]. In **Figure 9**, the photonic efficiency has increased significantly due to the doping of these metallic elements. Although the type of doped element does not make a significant difference, the highest efficiency is related to the nanolayer doped with gold, followed by silver. It can be seen from **Figure 6** that the longer the samples are exposed to UV radiation, the more the photonic efficiency decreases. This occurs for all samples, but with slightly different attenuation coefficients and the photonic efficiency decreases almost exponentially with time. This effect is due to the recombination of electron-hole pairs and the reduction of photocatalytically active sites, which is an exponential function of time [45,46].

4. Conclusion

In the presence of metallic contaminants such as copper, silver, and gold, the surface roughness of the nanolayers increased, which increased the number of surface active sites on the photocatalyst surface, resulting in an increase in the photonic efficiency and absorption coefficient of the samples. The absorption coefficients of pure titanium dioxide nanolayers and those doped with copper, silver, and gold nanoparticles sputtered onto quartz substrates are all in the ultraviolet range and above. However, doping these metal elements in the titanium dioxide nanolayer structure causes a shift in the absorption spectrum and also the absorption edge of the titanium dioxide nanolayer towards the optical region of the electromagnetic spectrum. The magnitude of this shift in the absorption spectrum and absorption edge in the gold-doped titanium dioxide nanolayer is slightly greater than that in the silver-doped titanium dioxide nanolayer, and the silver-doped titanium dioxide nanolayer is slightly greater than that in the copper-doped titanium dioxide nanolayer, which can be attributed to the change in the size of the doped elements and a slight difference in the location of the Fermi level of these three elements.

Titanium dioxide nanolayers doped with copper, silver, and gold elements also have better photocatalytic activity and higher photonic efficiency, which is due to the difference in the size of the impurities, which creates a larger effective surface area and increases the number of active sites in the photocatalytic activity, and the second effect is to reduce the electron-hole pair recombination process in the photocatalytic activity by trapping surface electrons by the metal. In other words, while the metal species increase the number of active sites in the photocatalytic activity, they also act as electron trapping sites. By spatially separating the light-generated electrons from the holes, this trapping effectively reduces the rate of harmful electron-hole recombination, thereby maximizing the number of carriers available for the photocatalytic reaction. The slight difference observed in the doped samples is due to this difference in the location of their Fermi level, which is located in the band gap of titanium dioxide. Therefore, titanium dioxide nanolayers doped with gold, silver and copper elements, respectively and with a slight difference, have the highest photon efficiency in the photocatalytic activity.

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Conflict of interest: The author declare they have no competing interests.

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